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希土類珪酸塩ガラスの作製と相転移挙動の観察

Fabrication of rare earth silicate glasses and observation
of phase transition behavior

大川采久¹, Nguyen Thanh Son², 長谷川拓哉¹, 中山忠親³, 石川毅彦⁴, 下西里奈⁴, 小山千尋⁴, 織田裕久⁴, Do Thi Mai Dung³, 殷澍^{1,5}
Ayahisa OKAWA¹, Nguyen Thanh SON², Takuya HASEGAWA¹, Tadachika NAKAYAMA³, Chihiro KOYAMA⁴, Hirohisa ODA⁴, Rina Shimonishi⁴, Takehiko ISHIKAWA⁴, Do Thi Mai Dung³ and Shu YIN^{1,5}

¹ 東北大学 多元物質科学研究所, Institute of Multidisciplinary Research for Advanced Material (IMRAM), Tohoku University

² 釧路工業高等専門学校 創造工学科, Department of Creative Engineering, National Institute of Technology, Kushiro College

³ 長岡技術科学大学 極限エネルギー密度工学研究センター, Extreme Energy-Density Research Institute, Nagaoka University of Technology

⁴ 宇宙航空研究開発機構, Japan Aerospace Exploration Agency, Tsukuba

⁵ 東北大学 材料科学高等研究所, Advanced Institute for Materials Research (WPI-AIMR), Tohoku University

1. Introduction

SiC fiber reinforced SiC matrix composites ($\text{SiC}_f/\text{SiC}_m$) demonstrate potential as the next generation of gas turbine blades. However, SiC exhibits recession at high temperatures in an atmosphere with high water vapor. Therefore, an environmental barrier coating (EBC) is required. A general EBC system includes a Si bond coat, which obstructs oxidation and enhances adhesion, and a top coat of rare earth (RE) silicates such as RE_2SiO_5 and $\text{RE}_2\text{Si}_2\text{O}_7$. Recently, high entropy RE silicates with five or more equivalent molar RE have garnered attention to improve the properties for EBC. The most promising structure is the polymorphism-free $\beta\text{-RE}_2\text{Si}_2\text{O}_7$ structure (with 6-coordination of RE less than 0.885 \AA)¹.

To prevent water vapor recession of SiC, the EBC must be dense. However, coatings fabricated by typical thermal spraying processes are rapidly cooled and amorphized by the temperature difference with the substrate. Subsequent annealing crystallizes the coating, but eventually cracks and pores remain, compromising the effectiveness of the EBC. Although an irreversible phase transition from a metastable phase to a stable phase with volume expansion during annealing is attractive to cause crack healing in coatings², an understanding of the thermal evolution of the coating microstructure and phases has not been established. The analysis is complicated by the volatilization of SiO in thermal spraying, the local element segregation of the coatings³, and the multi-element composition of RE⁴. In addition, the thermophysical properties which are important parameters in thermal spraying⁵, have not been reported for RE silicate. The purpose of this study is to reveal the thermal evolution of microstructure and phases in RE silicate glasses without local element segregation, thereby demonstrating crack healing, and to measure the thermophysical properties.

2. Experimental Procedures

The RE elements Lu, Yb, Er, Y, and Ho were chosen to form β -RE₂Si₂O₇. Henceforth, RE sites with equal molar ratios of Lu, Yb, and Er will be denoted as 3RE, and when Ho and Y are included, it will be represented as 5RE. RE nitrate (RE(NO₃)₃ · nH₂O) and sodium silicate (Na₂SiO₃ · 9H₂O) were dissolved in 30 mL of distilled water, respectively. The solutions were prepared to have stoichiometric ratios of RE:Si = 0.5 and 1.0. 2M HCl was added to the Na₂SiO₃ solution, while diluted ammonia was added dropwise to the RE(NO₃)₃ solutions. The two solvents were mixed, followed by sealing the mixed solution in a 100-mL autoclave with a 70% fill rate and holding at 220°C for 3 h. After washing by centrifugation, the obtained powder was exposed at 1600°C for 10 h in air. The powder was irradiated with a CO₂ laser using a gas-levitation furnace. The crystallographic phases of the powders before irradiated with a CO₂ laser were analyzed using X-ray diffraction. The CO₂ laser-irradiated samples were measured by Raman spectroscopy.

3. Results and discussion

From the XRD patterns, the main phases of the synthesized RE₂Si₂O₇ and RE₂SiO₅ powders were identified as β -RE₂Si₂O₇ and X2-RE₂SiO₅, respectively. The diffraction peaks are slightly shifted toward the lower diffraction angles as the average RE ionic radius becomes larger than that of Yb as the RE element increases, causing lattice expansion. Figure 1 shows the photograph of CO₂ laser irradiated samples. RE₂SiO₅ showed recalescence after melting and quenching. In addition, RE₂SiO₅ was non-transparent, which suggests crystallization. Whereas RE₂Si₂O₇, with a SiO₂-rich composition, was transparent and vitrified, and pores were observed in the appearance of RE₂Si₂O₇. The Raman spectrum of RE₂SiO₅, the increase in the RE elements causes a marked broadening of the peak widths, indicating the presence of disordered atomic bonding. In addition, the Raman peaks corresponding to the Si-O bonds also become broader, which is attributed to the lattice distortion of the Si- and O- sublattices⁶. On the other hand, glassy RE₂Si₂O₇ shows only broad bands, suggesting that it was amorphous, similar to the as-sprayed RE₂Si₂O₇⁷.

Figure 1. Photograph of the appearance of RE silicate after irradiation with a CO₂ laser

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