

## P19

## 六方晶 $\text{DyMnO}_3$ 相における雰囲気酸素分圧とトレランス因子( $TF$ )の関係

### Relationship between ambient oxygen partial pressure and tolerance factor ( $TF$ ) in hexagonal $\text{DyMnO}_3$ phase

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#### 1. Introduction

Hexagonal  $\text{DyMnO}_3$  phase ( $h\text{-DyMnO}_3$ , space group:  $P6_3cm$ ) is known as a functional material because of the presence of ferroelectricity and (anti) ferromagnetism in one phase. The structural stability of  $h\text{-DyMnO}_3$  can be evaluated by the following tolerance factor ( $TF$ ) (1) using ionic radius of composed element,  $r_i$ .

$$TF = \frac{r_{\text{Dy}} + r_{\text{O}}}{\sqrt{2}(r_{\text{Mn}} + r_{\text{O}})} \quad (1)$$

It has been reported that the reduction in valence from  $\text{Mn}^{3+}$  to  $\text{Mn}^{2+}$  results in oxygen deficiency followed by a formation of  $h\text{-DyMnO}_3$  when ambient oxygen partial pressure ( $P_{\text{O}_2}$ ) is low<sup>1,2</sup>, though stable phase of  $\text{DyMnO}_3$  is orthorhombic  $\text{DyMnO}_3$  ( $o\text{-DyMnO}_3$ , space group:  $Pbnm$ )<sup>3</sup>.

In this study, we aimed to clarify the relationship between the  $TF$  and the reason why  $h\text{-DyMnO}_3$  tends to be formed when the oxygen partial pressure is decreased.

#### 2. Experimental procedure

Powders of  $\text{Dy}_2\text{O}_3$  and  $\text{Mn}_2\text{O}_3$  mixed in an agate mortar were melted on a copper hearth by using a semiconductor laser to form a spherical  $\text{DyMnO}_3$  sample with a diameter of 2 mm. The spherical sample was melted and then solidified from the undercooled melt at levitation state by an aerodynamic levitation furnace using argon and  $\text{O}_2$  gases. The temperature and solidification process of the sample was recorded by a monochromatic pyrometer and a high-speed video camera (HSV), respectively. The solidified samples were analyzed by powder X-ray diffractometer (XRD), scanning electron microscope (SEM) and thermogravimetric differential scanning calorimeter (TG-DTA).

#### 3. Results

**Figure 1** shows a typical example of the results of TG-DTA operated under high purity oxygen gas at  $P_{\text{O}_2} = 10^5$  Pa an as-solidified  $\text{DyMnO}_3$  sample. The weight of the sample increases from at about 500 K and saturates at about 1200 K. The weight increment during heating is regarded as an increase in oxygen due to oxidation of  $\text{Mn}^{2+}$  to  $\text{Mn}^{3+}$ , corresponding to the amount of the oxygen deficiency. Assuming that the ratio of the radius between the oxygen vacancy and the oxygen ion as 0.6, the effective oxygen ionic radius was calculated. Table 1 shows oxygen deficiency, effective ionic radii of Mn and oxygen, and calculated  $TF$  of  $\text{DyMnO}_3$ .

**Figure 2** shows the relationship between  $P_{\text{O}_2}$  and  $TF$  in  $\text{DyMnO}_3$ .  $TF$  changes in proportion to  $P_{\text{O}_2}$ . **Figure 3** shows the XRD profiles of  $\text{DyMnO}_3$  samples solidified at various  $P_{\text{O}_2}$ . Both  $h\text{-DyMnO}_3$  ( $\square$ ) and  $o\text{-DyMnO}_3$  ( $\triangle$ ) phases are identified in the sample solidified at  $P_{\text{O}_2} = 10^5$  Pa. However, the diffraction peaks for the  $o\text{-DyMnO}_3$  ( $\triangle$ ) phase disappears when the  $P_{\text{O}_2}$  is lower than  $3.2 \times 10^3$  Pa. Further decrease in  $P_{\text{O}_2}$  induces the formation of cubic  $\text{MnO}$  ( $c\text{-MnO}$ , space group:  $\text{Fm}\bar{3}\text{-m}$ ) ( $\circ$ ),

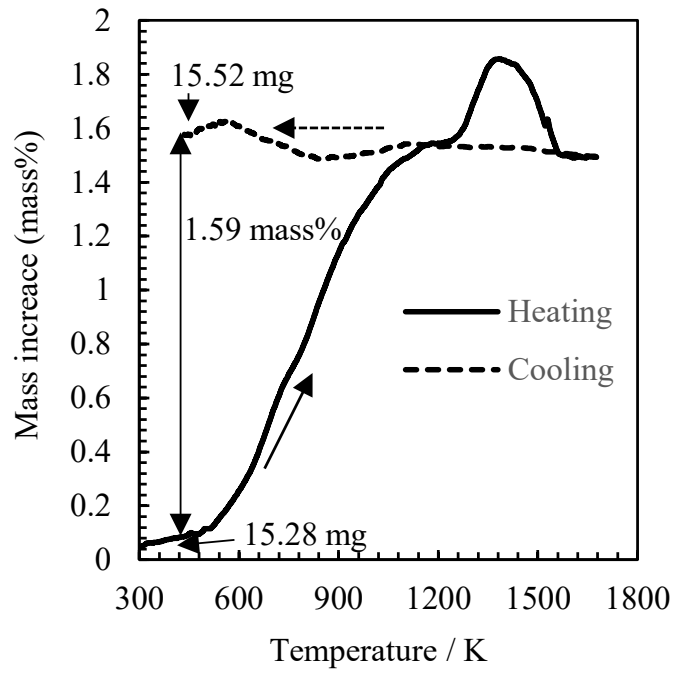


Fig. 1. TGA curve of the DyMnO<sub>3</sub> sample solidified at  $P_{O_2}$  of 10 Pa, carried out under high purity oxygen atmosphere

Table. 1. Oxygen deficiencies, average ionic radius, and  $TF$  of the as-solidified DyMnO<sub>3</sub> samples solidified under various  $P_{O_2}$

$P_{O_2}$ (Pa)	$1.0 \times 10^5$	$3.2 \times 10^3$	$1.0 \times 10^3$	10
<b>O<sub>2</sub> deficiency (%)</b>	0.2765	3.2020	4.5901	8.7931
<b>Average ionic radius (nm)</b>	<b>Mn</b>	0.0648	0.0681	0.0696
	<b>O<sub>2</sub></b>	0.1372	0.1355	0.1344
<b>Tolerance Factor</b>	0.7992	0.7875	0.7820	0.7654

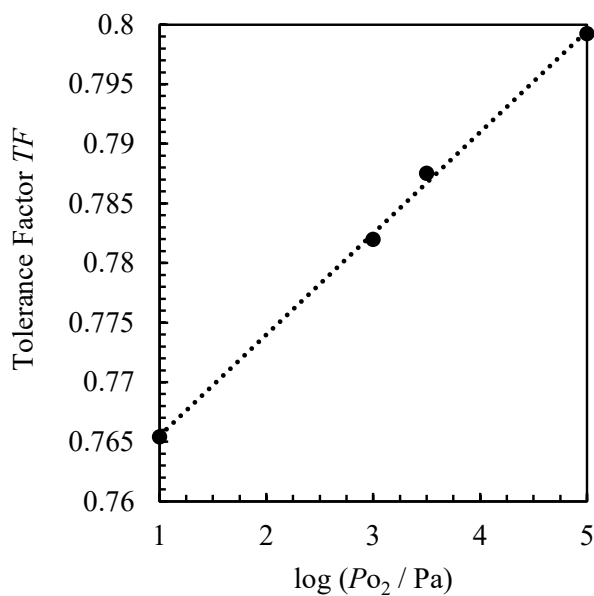


Fig. 2.  $TF$  of the as-solidified DyMnO<sub>3</sub> as a function of  $P_{O_2}$

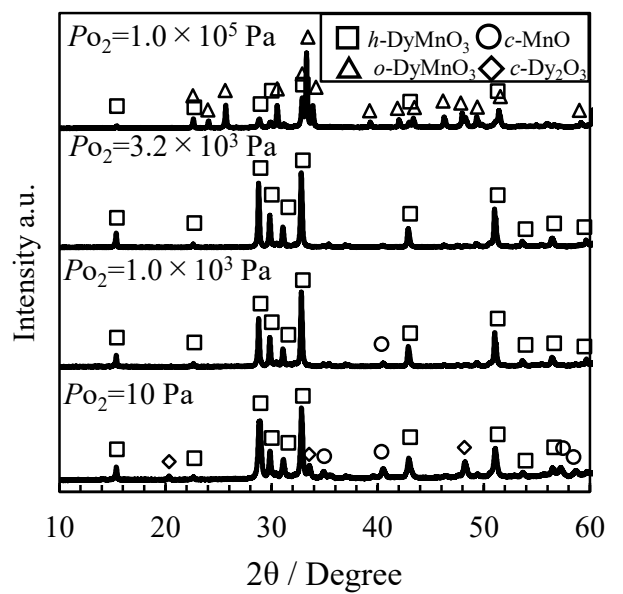


Fig. 3. XRD profiles of the as-solidified DyMnO<sub>3</sub>

which would be attributed by the reduction in valence from  $Mn^{3+}$  to  $Mn^{2+}$ . When the  $P_{O_2}$  is decreased to  $10^1$  Pa, cubic  $Dy_2O_3$  ( $c$ - $Dy_2O_3$ , space group:  $1a-3$ ) ( $\diamond$ ) is formed together with the  $h$ - $DyMnO_3$  and  $c$ - $MnO$ .

## Reference

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